

## Microporous Organic Nanoparticles Anchoring CeO<sub>2</sub> Materials: Reduced Toxicity and Efficient Reactive Oxygen Species-Scavenging for Regenerative Wound Healing

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**Abstract:** This work shows that microporous organic polymer can be applied to the engineering of  $CeO_2$ -based antioxidant nanomaterials for *in vivo* regenerative wound healing. Whilst nanoparticulate  $CeO_2$  has shown excellent redox quenching of reactive oxygen species (ROS) because it is rich in oxygen defects and  $Ce^{3+}$  species, it has a strong tendency to form aggregates because it is a kinetic intermediate of bulk materials. Anchoring of nanoparticulate  $CeO_2$  on suitable

#### Introduction

The understanding, control, and utilization of reactive oxygen species (ROS) has been a continuing research subject.<sup>[1-2]</sup> Whilst the ROS are generated naturally in bio-systems,<sup>[1]</sup> they can be manipulated and controlled artificially.<sup>[2]</sup> For example, the ROS have been generated and utilized in the oxygen-based photo-catalytic decomposition of organic pollutants in water by inorganic semiconductor materials.<sup>[2]</sup>

In bio-systems, it has been known that injury of skin induces the generation of ROS in high amount.<sup>[3-4]</sup> At wound sites, the ROS are utilized in the oxidative defense against pathogens.

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supports can be an efficient anti-aggregation strategy. In this work,  $CeO_2$  nanoparticles were immobilized on microporous organic nanoparticles (MONPs) to form MONP@CeO\_2 nano-composites. Compared to  $CeO_2$  materials, the MONP@CeO\_2 nanocomposites showed not only efficient ROS scavenging in *in vivo* regenerative wound healing but also reduced cytotoxicity in *in vitro* cell studies, due to the efficient distribution and dilution effect of CeO<sub>2</sub> nanomaterials.

Although the ROS play a constructive role in wound healing, usually, over-generated ROS are problematic and damage tissues. Recently, the scavenging of ROS has attracted significant attention of scientists as a strategy to help regenerative wound healing.<sup>[3]</sup> In this regard, recently, functional additives have been developed for the ROS scavenging in wound sites.<sup>[1,3,5]</sup>

Recently, CeO<sub>2</sub> nanomaterials have attracted special attention of scientists in bio-research field.<sup>[5]</sup> Among inorganic materials, the nanoparticulate CeO<sub>2</sub> has shown excellent redox quenching of ROS because it is rich in oxygen vacancy and Ce<sup>3+</sup> species.<sup>[5-6]</sup> Because the oxygen vacancy and Ce<sup>3+</sup> species in CeO<sub>2</sub> increase sharply at nanoscale and thus, the ROS scavenging performance of CeO<sub>2</sub> is dependent on its size and surface area,<sup>[6]</sup> the size-maintenance of nanoparticulate CeO<sub>2</sub> is important for efficient ROS scavenging performance.

Recently, various inorganic materials have been engineered at nanoscale.<sup>[7]</sup> Whilst unprecedented chemical and physical properties have been discovered in the nano-scaled inorganic materials, the increased surface areas, due to reduced sizes, could enhance the functional performance of nanomaterials.<sup>[7–8]</sup> Nano-scaled inorganic materials are not thermodynamically stable and have a strong tendency to form bulk materials.<sup>[9]</sup> Thus, nano-scaled inorganic materials can be regarded as kinetically controlled systems.<sup>[10]</sup> Aggregation is an inevitable phenomenon of nanomaterials, implying that the benefits of their enhanced surface areas can easily be diminished.<sup>[11]</sup> To maintain the surface-related performance of nanomaterials, there should be efficient strategies to hinder the aggregation.<sup>[12]</sup> For example, anchoring of nanomaterials on solid supports can be an option to retard their aggregation.<sup>[13]</sup> For this, suitable chemistry for solid supports is required.

Recently, various microporous organic polymers (MOPs) have been prepared.<sup>[14]</sup> The size and morphology of MOP materials have been controlled to nanoscale.<sup>[15-16]</sup> It can be expected that the MOP nanoparticles (MONPs) can serve as

nanosupports for functional inorganic nanomaterials.<sup>[17]</sup> For example, the CeO<sub>2</sub> nanoparticles can be immobilized on MONP through the interaction of Ce species with  $\pi$ -electrons of MONP, hindering the aggregation of CeO<sub>2</sub> materials. However, as far as we are aware, the bioapplication of MOP is at an early stage.<sup>[18]</sup> Recently, our research group has reported the application of MOP to drug delivery systems for cancer therapy.<sup>[18e,f]</sup> Also, we have shown that organic polymers are helpful in the regenerative wound healing.<sup>[19]</sup> In this work, we report the synthesis of MONP@CeO<sub>2</sub> nanocomposites and their application to regenerative wound healing.

#### **Results and Discussion**

Figure 1 shows a synthetic scheme for MONP@CeO<sub>2</sub> nanocomposites. MONP nanosupports were prepared by Sonogashira coupling of tetra(4-ethynylphenyl)methane<sup>[20]</sup> with 1,4diiodobenzene in the presence of poly(vinylpyrrolidone) (PVP).<sup>[17]</sup> Then, CeO<sub>2</sub> was synthesized in the presence of MONP through the decomposition of Ce(OAc)<sub>3</sub>.<sup>[21]</sup> We gradually increased the amount of Ce(OAc)<sub>3</sub> from 0.079 mmol to 0.16, 0.24, and 0.32 mmol, resulting in four MONP@CeO<sub>2</sub> nanocomposites which were denoted as MONP@CeO<sub>2</sub>-1, MON-P@CeO<sub>2</sub>-2, MONP@CeO<sub>2</sub>-3, and MONP@CeO<sub>2</sub>-4, respectively. In addition, we prepared CeO<sub>2</sub> nanomaterials in the absence of MONP as a control material.

As shown Figure 2a, scanning electron microscopy (SEM) of the CeO<sub>2</sub> materials prepared in the absence of MONP showed granular aggregates with irregular sizes in the range of  $20 \sim$ 500 nm. Transmission electron microscopy (TEM) showed that



Figure 1. A synthetic scheme of MONP@CeO<sub>2</sub> nanocomposites.

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Figure 2. SEM images of (a) CeO<sub>2</sub>, (b) MONP, (c) MONP@CeO<sub>2</sub>-1, (d) MONP@CeO<sub>2</sub>-2, (e) MONP@CeO<sub>2</sub>-3, and (f) MONP@CeO<sub>2</sub>-4. TEM images of (g) CeO<sub>2</sub>, (h) MONP, (i) MONP@CeO<sub>2</sub>-1, (j) MONP@CeO<sub>2</sub>-2, (k) MONP@CeO<sub>2</sub>-3, and (l) MONP@CeO<sub>2</sub>-4.

the CeO<sub>2</sub> aggregates consist of small CeO<sub>2</sub> particles in the size range of 5 ~ 15 nm. (Figure 2g) SEM image of the MONP showed fairly homogeneous particles with an average size of 65 nm. (Figure 2b and S1 in the SI) TEM and high resolution (HR) TEM images showed microporosity of the MONP. (Figure 2h and S2 in the SI) SEM and TEM images of the MONP@CeO<sub>2</sub> showed that CeO<sub>2</sub> particles with sizes of  $3 \sim 5$  nm formed on the surface of MONP without independent CeO<sub>2</sub> materials. (Figures 2c–f, 2i–l, and S2 in the SI) The amount of CeO<sub>2</sub> nanoparticles on the MONP supports gradually increased from MONP@CeO<sub>2</sub>-1 to MONP@CeO<sub>2</sub>-2, MONP@CeO<sub>2</sub>-3, and MONP@CeO<sub>2</sub>-4. Elemental mapping analysis based on energy dispersive X-ray spectroscopy (EDS) supported the successful anchoring of CeO<sub>2</sub> nanoparticles on the MONPs (Figure S3 in the SI).

Powder X-ray diffraction (PXRD) studies showed that the MONP has amorphous characteristic<sup>[22]</sup> and that the nanoparticles on the MONP are CeO<sub>2</sub> (JCPDS # 81-0792) with (111), (200), (220), (311), (222), (440), and (331) diffraction peaks at  $2\theta$ of 28.6, 33.2, 47.4, 56.2, 59.0, 69.4, and 76.6°, respectively. (Figure 3a) X-ray photoelectron spectroscopy (XPS) of the CeO<sub>2</sub> and MONP@CeO<sub>2</sub> nanocomposites showed a mixed state of Ce<sup>3+</sup> (3d orbital peaks at 880.4, 884.9, 898.8, and 903.4 eV) and Ce<sup>4+</sup> species (3d orbital peaks at 882.1, 888.7, 898.1, 900.5,



Figure 3. (a) PXRD patterns and (b) XPS spectra of CeO<sub>2</sub>, MONP, and MONP@CeO<sub>2</sub> nanocomposites.

907.1, and 916.4 eV), which is a unique observation in the nanosized CeO<sub>2</sub> materials with oxygen vacancies<sup>[7]</sup> (Figure 3b).

By analysis of N<sub>2</sub> sorption isotherm curves, surface areas of the MONP and CeO<sub>2</sub> were measured to be 548 and 72  $m^2/q$ with pore volumes of 0.66 and 0.07 cm<sup>2</sup>/g, respectively. (Figure 4a) Surface areas of MONP@CeO2-1, MONP@CeO2-2, MON-P@CeO<sub>2</sub>-3, and MONP@CeO<sub>2</sub>-4 were measured to be 425, 315, 279, and 249 m<sup>2</sup>/g with pore volumes of 0.55, 0.51, 0.45, and 0.35 cm<sup>3</sup>/g, respectively (Figure 4b and Table 1).

In infrared (IR) absorption spectrum, the MONP showed main peaks at 1683, 1501, and 820 cm<sup>-1</sup>, corresponding to C=O (PVP), aromatic C=C, and aromatic C-H vibrations, respectively. (Figure 4c) In comparison, as the amount of CeO<sub>2</sub> in MON-P@CeO<sub>2</sub> increased, the intensities of IR peak at 507 cm<sup>-1</sup> (Ce–O vibration) increased in addition to the IR peaks of 1427, and 1548 cm<sup>-1</sup>, corresponding to the vibrations of surface oxygen species.<sup>[23]</sup> Thermogravimetric analysis (TGA) showed that the MONP and MONP@CeO2 nanocomposites are stable up to 325 °C. (Figure 4d) Solid state <sup>13</sup>C nuclear magnetic resonance (NMR) spectroscopy of MONP showed <sup>13</sup>C peaks at 121~145, 90, and 63 ppm, corresponding to aromatic groups, internal alkynes, and benzyl carbon, respectively. (Figure 4e) In addition, <sup>13</sup>C peaks were observed at 17~49 and 174 ppm, corresponding to aliphatic and carbonyl groups of PVP, respectively, supporting that the MONP with entrapped PVPs was formed by Sonogashira coupling of building blocks. The <sup>13</sup>C NMR spectrum of MONP@CeO<sub>2</sub>-2 was the same with that of MONP, indicating the retention of chemical structure (Figure S4 in the SI).

According to elemental analysis, the contents of CeO<sub>2</sub> in the MONP@CeO2-1, MONP@CeO2-2, MONP@CeO2-3, and MON-P@CeO<sub>2</sub>-4 were measured to be 18.7, 33.5, 46.3, and 54.9 wt%, respectively. (Table 1) The contents of Ce in the MONP@CeO<sub>2</sub>-1, MONP@CeO<sub>2</sub>-2, MONP@CeO<sub>2</sub>-3, and MONP@CeO<sub>2</sub>-4 were measured to be 1.45, 2.19, 2.84, and 2.86 mmol/g, respectively, by inductively coupled plasma (ICP) analysis.

Whilst zeta potentials of the CeO<sub>2</sub> and MONP were measured to be -14.5 and -14.9 mV in water, respectively, those of MONP@CeO2 composites were shifted to more negative values ( $-31.2 \sim -32.5$  mV), due to the efficient dis(a) (b) 400 0 <sup>300</sup> م <del>ک</del> 300 liff liff 3 4 5 6 e Size / nm V / cm<sup>3</sup> (STP) ( 00 00 ( cm<sup>3</sup> (STP) <u>6</u> MONP MONF @CeO--2 @CeO 5 100 CeO. MONE MONF @CeO-4 @CeO2-3 4 0.6 P/P 0.8 0.0 0.2 0.4 0.6 0.8 0.0 0.2 0.4 1.0 D/D (c) (d) MONF MONP 100 CeO, @CeO<sub>2</sub>-3 @CeO<sub>2</sub>-4 MONE % 80 MONP Weight Retention 6 0 MONF @CeO2-1 MON Intensity / a.u. @CeO<sub>2</sub>-1 @CeO2-2 MONF @CeO2-2 MONP @CeO2-3 20 MONP @CeO\_-4 2400 2000 1600 1200 800 200 400 ດ່າລ 800 Wavenumber / cm Temp. / °C (e) C3-4,9 C2 C10,14 C12 / C13 C15 C11 / 200 175 150 125 100 75 50 25 Chemical Shift / ppm

Figure 4. (a–b) N<sub>2</sub> adsorption-desorption isotherm curves at 77 K and pore size distribution diagrams (based on the DFT method), (c) IR spectra, and (d) TGA curves of CeO<sub>2</sub>, MONP, and MONP@CeO<sub>2</sub> composites. (e) Solid state <sup>13</sup>C NMR spectrum of MONP.

P@CeO <sub>2</sub> composites.							
Materials	CeO <sub>2</sub> <sup>[a]</sup> [wt%]	S <sub>BET</sub> <sup>[b]</sup> [m <sup>2</sup> /g]	V <sub>tot</sub> <sup>[c]</sup> [cm <sup>3</sup> /g]	<i>Z</i> <sup>[d]</sup> [mV]	D <sup>[e]</sup> [nm]		
CeO <sub>2</sub>	100	72	0.07	-14.5	255		
MONP	0	548	0.66	-14.9	83		
MONP@CeO <sub>2</sub> -1	18.7	425	0.55	-31.9	104		
MONP@CeO <sub>2</sub> -2	33.5	315	0.51	-32.5	126		
MONP@CeO2-3	46.3	279	0.45	-31.8	134		
MONP@CeO2-4	54.9	249	0.35	-31.2	135		
	Chemical and ph composites. Materials CeO <sub>2</sub> MONP MONP@CeO <sub>2</sub> -1 MONP@CeO <sub>2</sub> -2 MONP@CeO <sub>2</sub> -3 MONP@CeO <sub>2</sub> -4	Chemical and physical proposites.        Materials      CeO <sub>2</sub> <sup>[a]</sup> [wt%]        CeO <sub>2</sub> 100        MONP      0        MONP@CeO <sub>2</sub> -1      18.7        MONP@CeO <sub>2</sub> -2      33.5        MONP@CeO <sub>2</sub> -3      46.3        MONP@CeO <sub>2</sub> -4      54.9	Chemical and physical properties of 0 composites.        Materials      CeO <sub>2</sub> <sup>[a]</sup> [wt%]      S <sub>BFT</sub> <sup>[b]</sup> [m <sup>2</sup> /g]        CeO <sub>2</sub> 100      72        MONP      0      548        MONP@CeO <sub>2</sub> -1      18.7      425        MONP@CeO <sub>2</sub> -2      33.5      315        MONP@CeO <sub>2</sub> -3      46.3      279        MONP@CeO <sub>2</sub> -4      54.9      249	Chemical and physical properties of CeO <sub>2</sub> , MON        composites.      Materials      CeO <sub>2</sub> <sup>[a]</sup> [wt%]      S <sub>BET</sub> <sup>[b]</sup> [m <sup>2</sup> /g]      V <sub>tot</sub> <sup>[c]</sup> [cm <sup>3</sup> /g]        CeO <sub>2</sub> 100      72      0.07        MONP      0      548      0.66        MONP@CeO <sub>2</sub> -1      18.7      425      0.55        MONP@CeO <sub>2</sub> -2      33.5      315      0.51        MONP@CeO <sub>2</sub> -3      46.3      279      0.45        MONP@CeO <sub>2</sub> -4      54.9      249      0.35	Chemical and physical properties of CeO2, MONP, and MC composites.      Materials $CeO_2^{[a]}$ $S_{BET}^{[b]}$ $V_{tot}^{[c]}$ $Z^{[d]}$ $[mV]$ CeO2    100    72    0.07    -14.5      MONP    0    548    0.66    -14.9      MONP@CeO2-1    18.7    425    0.55    -31.9      MONP@CeO2-2    33.5    315    0.51    -32.5      MONP@CeO2-3    46.3    279    0.45    -31.8      MONP@CeO2-4    54.9    249    0.35    -31.2		

[a] Contents of CeO<sub>2</sub> based on elemental analysis. [b] Surface areas based on the Brunauer-Emmett-Teller theory. [c] Total pore volume. [d] Zeta potential in water. [e] Hydrodynamic diameter measured by DLS studies in water.

tribution of CeO<sub>2</sub> on the MONP. (Table 1 and Figure S5 in the SI) The existence of PVP made the MONP water-compatible.<sup>[17]</sup> Whilst hydrodynamic average diameters of the CeO<sub>2</sub> and MONP were measured to be 255 and 83 nm, respectively, by dynamic light scattering (DLS) studies, those of MONP@CeO<sub>2</sub> composites increased gradually from 104 nm (MONP@CeO<sub>2</sub>-1) to 126 (MONP@CeO2-2), 134 (MONP@CeO2-3), and 135 nm (MON-P@CeO<sub>2</sub>-4) with an increase of CeO<sub>2</sub> amount. (Table 1 and

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ChemNanoMat 2020, 6, 1-8 www.chemnanomat.org These are not the final page numbers! 77 Figure 5a) It is noteworthy that the elimination process of nanomaterials from body depends on their sizes.<sup>[24-25]</sup> It is well known that the nanomaterials with sizes greater than 10 nm extravasate into liver through liver sinusoidal fenestrations and thereafter, hepatocytes in the liver eliminate foreign substances by endocytosis, followed by enzymatic breakdown and excretion into the bile.<sup>[24-25]</sup>

To study ROS scavenging function of the MONP@CeO2 composites, we tested their catalytic performance in the  $H_2O_2$ decomposition to water and O<sub>2</sub>.<sup>[26]</sup> (Figure 5b) It is noteworthy that H<sub>2</sub>O<sub>2</sub> is one of main ROS generated in wound sites.<sup>[1–5]</sup> The control CeO<sub>2</sub> material showed a relatively poor catalytic performance, due to its aggregated nature. (Figure 5b) Whilst the MONP@CeO<sub>2</sub>-1 showed good catalytic performance over 3 h, its catalytic durability decreased after 3 h, implying that the proper aggregation of CeO<sub>2</sub> nanoparticles enhances the stability of systems. (Figure 5b) After 8 h, the catalytic efficiencies of  $CeO_2$  in the materials were in the order of MONP@CeO<sub>2</sub>-2>  $MONP@CeO_2-3 > MONP@CeO_2-4 > MONP@CeO_2-1 > CeO_2$ , indicating that the MONP serves as a good nanosupport for the efficient distribution of CeO<sub>2</sub> materials. In a control test, the MONP showed negligible catalytic activity in the H<sub>2</sub>O<sub>2</sub> decomposition (Figure S6 in the SI).

In vitro cytotoxicity of materials toward human 293T and mouse NIH 3T3 cells was investigated by MTT assay. (Figures 5c–d and S7–8 in the SI). whilst the  $CeO_2$  showed potential cytotoxicity (cell viability <90%) in the concentration of



**Figure 5.** (a) Hydrodynamic diameters of CeO<sub>2</sub>, MONP, and MONP@CeO<sub>2</sub> nanocomposites in water. (b)  $H_2O_2$  decomposition to water and  $O_2$  catalyzed by CeO<sub>2</sub> and MONP@CeO<sub>2</sub> nanocomposites. (refer to Figure S5 in the SI for the catalytic activity of MONP for the  $H_2O_2$  decomposition). *In vitro* cytotoxcity of (c) CeO<sub>2</sub> and (d) MONP@CeO<sub>2</sub> nanocomposites measured by MTT assay using human embryonic kidney 293T cells (refer to Figure S6 in the SI for MTT assay of MONP).

 $>10~\mu g/mL,~cytotoxicity~was~significantly~reduced in the MONP@CeO_2 composites with cell viabilities <math display="inline">>90\%$  even at 1000  $\mu g/mL$ , due to the dilution effect of the CeO\_2 in the MONP materials. (Figures 5c–d) In a control test, the MONP showed cell viability of 86% at 1000  $\mu g/mL$  (Figure S7 in the SI).

Next, we studied the effects of CeO<sub>2</sub>, MONP, and MON-P@CeO<sub>2</sub> materials in *in vivo* regenerative wound healing of Spraque-Dawley (SD) rats (n = 3). The same amount of CeO<sub>2</sub>, MONP, and MONP@CeO<sub>2</sub> nanocomposites were added to linear cuts (1 cm length and 1~1.2 mm depth, penetrating a subcutaneous tissue layer<sup>[27]</sup> of skin) of SD rats. (Figure 6a) The closure kinetics of cuts was investigated and quantificated (n = 3) by measuring the changes of wounds. Figures 6b–7 summarize the results. Whilst all MONP@CeO<sub>2</sub> nanocomposites showed better performance than a blank control system, CeO<sub>2</sub>, and MONP, the MONP@CeO<sub>2</sub>-2 showed the best performance. In addition, as time passed, the beneficial effect of CeO<sub>2</sub> distribution on the MONP became more significant (Figure 6b).

After 7 days, whilst a blank control system showed wound retention of  $43\pm3\%$ , the control materials of CeO<sub>2</sub> and MONP showed wound retention of  $26\pm3$  and  $27\pm2\%$ , respectively. (Figure 7a) The enhanced wound healing by CeO<sub>2</sub> and MONP, compared to the blank control system, is attributable to the ROS scavenging function of CeO<sub>2</sub> and the nanobridging effect<sup>[19]</sup> of nanoadditives in wounds, as illustrated in Figure 6a.

In cases of the MONP@CeO<sub>2</sub> nanocomposites, after 7 days, whilst the MONP@CeO<sub>2</sub>-1, MONP@CeO<sub>2</sub>-3, and MONP@CeO<sub>2</sub>-4 showed much enhanced wound healing with wound retention of  $18\pm2$ ,  $14\pm2$  and  $22\pm2\%$ , respectively, the MONP@CeO<sub>2</sub>-2 showed nearly complete wound healing with wound retention of  $0\pm1\%$ . (Figure 7b) The observed results are attributable to the optimal distribution of CeO<sub>2</sub> by MONP in the MONP@CeO<sub>2</sub>-2 nanocomposite, inducing the efficient redox quenching of ROS and the reduced cytotoxicity due to the dilution effect of CeO<sub>2</sub>. In addition, the enhanced negative zeta potentials of MON-P@CeO<sub>2</sub> nanocomposites, compared to CeO<sub>2</sub>, can be beneficial in maintaining their ROS scavenging functions. We suggest that the contamination of nanocomposites by blood proteins such as albumin with negative zeta potentials can be suppressed through charge-charge repulsion.<sup>[28]</sup>

After regenerative wound healing for 7 days, the resulant skin tissues were investigated by histological analysis. As shown in Figure 7c, the H&E and Masson's trichrome-stained skin tissues indicate complete re-epithelialization and recovery of collagen structures after wound healing by the MONP@CeO<sub>2</sub>-2 for 7 days.

#### Conclusion

This work shows that MOP nanoparticles can serve as nanosupports of functional inorganic materials for bioapplications. The ROS scavenging function of  $CeO_2$  materials could be enhanced through the distribution of  $CeO_2$  materials on the MONP. In addition to the anti-aggregation benefit, the MON-P@CeO<sub>2</sub> nanocomposites showed reduced cytotoxicity, compared to CeO<sub>2</sub>, due to the dilution effect. The MONP@CeO<sub>2</sub>-2



**Figure 6.** (a) A schematic illustration of the MONP@CeO<sub>2</sub>-assisted regenerative wound healing process. (b) Photographs of the *in vivo* regenerative wound healing processes of Spraque-Dawley rats with linear cuts (1 cm length and 1~1.2 mm depth) over 7 days by a blank control system, CeO<sub>2</sub>, MONP, and MONP@CeO<sub>2</sub> nanocomposites.



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**Figure 7.** (a–b) Quantificated *in vivo* wound closure kinetics of CeO<sub>2</sub>, MONP, and MONP@CeO<sub>2</sub> nanocomposites (SD rats: n = 3). (c) H&E and Masson's trichrome stained skin tissues (SD rats) after regenerative wound healing by a blank control system and MONP@CeO<sub>2</sub>-2 for 7 days.

with an optimal composition showed the best performance. We believe that the MOP nanoparticles with high surface areas and microporosity can be applied to the engineering of more various nanocomposites for bioapplications.

## **Experimental Section**

#### **Generation information**

SEM and TEM images were obtained by a FE-SEM (JSM6700F) and a JEOL 2100F, respectively. PXRD patterns were obtained by a Rigaku MAX-2200.  $N_2$  adsorption-desorption isotherm curves were obtained by a Micromeritics ASAP2020. Pore size distribution was analyzed by the DFT method. IR absorption spectra were obtained by a Bruker VERTEX 70 FT-IR spectrometer. TGA curves were obtained by a Seiko Exstar 7300. Solid state <sup>13</sup>C NMR spectrum was obtained at CP/TOSS mode by a 500 MHz Bruker ADVANCE II NMR spectrometer at the NCIRF of Seoul National University. Elemental analysis was performed by a CE EA1110 analyzer. ICP analysis was conducted using an OPTIMA 8300. Zeta potentials and hydrodynamic diameters were measured by a Zetasizer ZS90 (Malvern). XPS spectra were obtained by a Thermo VG spectrometer.

#### Synthesis of MONP, CeO<sub>2</sub>, and MONP@CeO<sub>2</sub> nanocomposites

MONP was prepared by the synthetic procedures modified from the literature.<sup>[17]</sup> In this study, for the preparation of MONP, (PPh<sub>3</sub>)<sub>2</sub>PdCl<sub>2</sub> (21 mg, 30 µmol), Cul (5.8 mg, 30 µmol), and triethylamine (25 mL) were added to a 100 mL Schlenk flask under argon. Polyvinylpyrrolidone (PVP, Mw: 40000, 1.55 g) dissolved in ethanol (50 mL) was added to the reaction mixture. After the reaction mixture was sonicated for 40 min at room temperature, tetra(4-ethynylphenyl)methane<sup>[20]</sup> (0.125 g, 0.300 mmol) and 1,4-diiodobenzene (0.198 g, 0.600 mmol) were added. The reaction mixture was stirred at 100 °C for 1 day. After being cooled to room temperature, solid was separated by centrifugation, washed with methanol (40 mL) thrice and methylene chloride (40 mL) thrice, and dried under vacuum.

For the preparation of MONP@CeO<sub>2</sub>-2, MONP (25 mg) was added to ethanol (5 mL) in a 25 mL Schlenk flask. The reaction mixture was sonicated for 15 min at room temperature. Cerium acetate hydrate (50 mg, 0.16 mmol) was dissolved in distilled water (20 mL) through sonication in a vial. After the cerium acetate aqueous solution was added, the reaction mixture was stirred at 80 °C for 1 day. The solid was separated by centrifugation, washed with water (40 mL) thrice and ethanol (40 mL) thrice, and dried under vacuum. For the preparation of MONP@CeO<sub>2</sub>-1, MONP@CeO<sub>2</sub>-3, and MONP@CeO<sub>2</sub>-4, 25 mg (0.079 mmol), 75 mg (0.24 mmol), and 100 mg (0.32 mmol) of cerium acetate hydrate were used. The other synthetic procedures were the same as those applied for MONP@CeO<sub>2</sub>-1. For the preparation of a control CeO<sub>2</sub> nanomaterial, the same synthetic procedures as those applied for MONP@CeO<sub>2</sub>-4 were applied without using MONP.

## Experimental procedures for the catalytic decomposition of $\rm H_2O_2$

For the catalytic decomposition of H<sub>2</sub>O<sub>2</sub> by CeO<sub>2</sub> and MONP@CeO<sub>2</sub> nanocomposites, the experimental procedures modified from the literature<sup>[29]</sup> were applied. A separatory funnel with water was connected to one neck of a 100 mL two-neck Schlenk flask using a silicon tube and a glass joint. A gas bubbler was connected to a cork of the 100 mL two-neck Schlenk flask using a silicon tube. After  $H_2O_2$  solution (0.20 M, 25 mL) was added to catalytic material (20 mg) in the 100 mL two-neck Schlenk flask at 25 °C (water bath temperature), the other neck was closed using a rubber septum. The reaction mixture was stirred at 300 rpm. As the oxygen gas was generated, the meniscus of the gas bubbler went down. To maintain the original height of the meniscus, the cork of the separatory funnel was opened. The volume of drained water at the given reaction time was measured by a messcylinder. Considering the vapor pressure of water at 25 °C, the volume of the generated oxygen gas was calculated.

#### In vitro cytoxicity

The In vitro cytotoxicity of MONP, CeO<sub>2</sub>, and MONP@CeO<sub>2</sub> nanocomposites was examined using human embryonic kidney 293T and mouse NIH 3T3 cells via MTT assay. The cells were cultured in RPMI medium supplemented with 10% FBS and 1% pencillinstreptomycin in a humidifed atmosphere. Thereafter, the cells were trypsinized, seeded in a 96-well plate at a density of  $1 \times 10^4$  cells/ well and allowed to grow for 24 h. The cells were then washed with PBS and incubated with various concentrations of MONP, CeO<sub>2</sub>, and MONP@CeO2 nanocomposites for 24 h. The cells incubated with medium alone served as a control system. Then, the cells were washed twice with PBS to remove remaining materials and incubated with fresh medium containing MTT solution (20 µL, 5 mg/mL) for 4 h. The culture medium was then removed and the purple crystals (formazan) were dissolved uisng DMSO to examine the viability. The absorbance of each well (at 490 nm) was measured on a microplate reader (Multiskan Go, Winooski, VT, USA) to calculate the cell viability.

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#### Wound healing

For our wound healing studies, Sprague-Dawley rats (SD rats, 200– 220 g) were purchased from the Korea Research Institute of Bioscience and Biotechnology (KRIBB, Daejeon, Korea). Live animal experiments were performed in accordance with the institutional guidelines of Sungkyunkwan University. The Sungkyunkwan University institutional committees approved all of our animal experiments.

For the wound healing studies, the SD rats were anesthetized with pentobarbital and then, linear open wounds (1 cm length and 1~ 1.2 mm depth) were created. The SD rats were randomly divided into seven different groups (n=3); (i) a blank control system (without any treatment), (ii) CeO<sub>2</sub> (5 mg/ml), (iii) MONP (5 mg/ml), (iv) MONP@CeO<sub>2</sub>-1 (5 mg/ml), (v) MONP@CeO<sub>2</sub>-2 (5 mg/ml), (vi) MONP@CeO<sub>2</sub>-3 (5 mg/ml), (vii) MONP@CeO<sub>2</sub>-4 (5 mg/ml). The hyaluronic acid (HA, 10 mg/ml) was added to the CeO<sub>2</sub>, MONP, MONP@CeO<sub>2</sub> nanocomposite solutions. The materials were well dispersed in aqueous solution and stayed on the wound bed. After the wound creation, no treatments were provided for the blank control group. For other cases, the CeO<sub>2</sub>, MONP, and nanocomposite solutions (150 µL) were transferred to small syringe (1 mL, 26G) and added to the wound area. The wound healing was monitored at 0, 1, 2, 3, 5, and 7 days and photographed to measure the extent of wound healing. Statistical analysis between the control and other tests was performed by one-way analysis of variance (ANOVA). Values indicate a mean standard deviation.

After 7 days, the rats were euthanized and skins were harvested. For histological analysis, the collected skin tissues were fixed using 10% formalin and embedded in paraffin. Then, skin tissues were sliced (~4  $\mu$ m thickness) and used for hematoxylin & eosin (H&E) and Masson's trichrome (MT) staining. The stained slides were visualized using the microscope to examine the order of wound healing.

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## **Conflict of Interest**

The authors declare no conflict of interest.

Keywords: Microporous organic polymer  $\cdot$  CeO\_2  $\cdot$  Nanocomposites  $\cdot$  Reactive oxygen species  $\cdot$  Regenerative wound healing

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## **FULL PAPER**

**Microporous organic nanoparticle** (**MONP**)@**CeO**<sub>2</sub> **nanocomposites** were engineered by the formation of CeO<sub>2</sub> materials in the presence of MONP. The amount of CeO<sub>2</sub> in the MONP@CeO<sub>2</sub> nanocomposites was optimized by the systematic control of Ce precursors. Compared to CeO<sub>2</sub> materials, the MONP@CeO<sub>2</sub> nanocomposites enhanced *in vivo* regenerative wound healing due to efficient scavenging of ROS and reduced cytotoxicity.



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Microporous Organic Nanoparticles Anchoring CeO<sub>2</sub> Materials: Reduced Toxicity and Efficient Reactive Oxygen Species-Scavenging for Regenerative Wound Healing