

Photocatalytic Effect of Ag/TiO₂ Nanotubes Fabricated Using 40 nm-Scale BCP Lithography

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Uniform TiO₂ nanotubes were fabricated using a self-aligned, hole patterned BCP template lithography, atomic layer deposition of TiO₂, and reactive ion etching. Also, Ag nanoparticles were deposited onto TiO₂ nanotubes using a photochemical deposition method. Using the TiO₂ nanotubes deposited with and without Ag nanoparticles, the effect of TiO₂ nanotubes and Ag nanoparticles on the photocatalytic degradation was investigated by measuring the concentration change of the methylene blue solution under a UV light irradiation. Anatase phase TiO₂ nanotubes with 50~60 nm height and 40~50 nm diameter could be successfully fabricated and Ag nanoparticles with 7~9 nm size could be also formed on the TiO₂ nanotubes. The use of TiO₂ nanotubes instead of flat TiO₂ film improved the photocatalytic effect due to the increased TiO₂ surface area and also the use of Ag nanoparticles on TiO₂ nanotubes further improved the photocatalytic effect up to 8% by slowing down the rate of electrons and holes recombination on the surface of TiO₂. However, when the coverage of Ag nanoparticles on TiO₂ nanotubes are too high, possibly due to the decreased UV penetration to TiO₂ by the increased surface area covered with Ag nanoparticles, the photocatalytic effect was decreased.

Keywords: TiO₂ Nanotubes, Environmental Photocatalysts, Ag Nanoparticles, Reactive Ion Etching.

1. INTRODUCTION

Titanium dioxide (TiO₂) is one of the multipurpose materials used for various functional applications such as solar cells, antibacterial agents, and photocatalysts for degradation of environmental contaminants due to its unique properties.^{1–7} Especially, the environmental decontamination by nanosize TiO₂ photocatalysis is investigated to be appealing because of its low cost, nontoxicity, high surface-to-volume ratios, high surface activity, and simple recycling.^{8,9}

The brief mechanism of the TiO₂ photocatalytic process is summarized in Figure 1.^{10–12} When TiO₂ in an aqueous dye solution is irradiated with UV or sun light, where the energy is greater than its band gap energy, two types of carriers, electrons and holes, are photogenerated and separated. The generated electrons from the valance band to the conduction band can react with O₂ or acceptor molecule A. The reacted O₂ can be reduced to superoxide anion O₂^{-•} and acceptor molecule can be reduced A^{-•}. The separated holes oxidize water or OH⁻ to produce OH[•] radicals through a one-electron oxidation step. The OH free radicals and the superoxide anions degrade the dye compounds.

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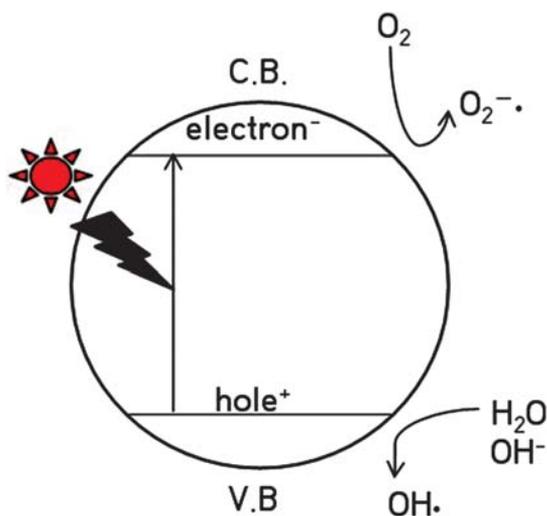


Fig. 1. Brief mechanism of the photocatalytic process of TiO₂.

Typically, two structured TiO₂ are being used, rutile and anatase.^{1,2,4,13,14} Despite having the same chemical composition, due to the different chemical bonding of rutile and anatase structured TiO₂, they have different band gap energies of 3.03 eV and 3.20 eV, respectively. Anatase structured TiO₂ exhibits a significantly higher photocatalytic activity than rutile structured TiO₂ because of the higher charge-carrier mobility of 80 cm²/V · s for anatase TiO₂, which is about 89 times faster than that of rutile TiO₂. Thus, anatase TiO₂ has better photocatalytic properties than rutile TiO₂. Also, at bulk phase, rutile TiO₂ is more stable while anatase TiO₂ is more stable when the TiO₂ size is less than 15 nm such as nanoparticles and nanotubes.

However, for nanosize anatase TiO₂, the separated electrons and holes are easily recombined, too, due to higher defect density at the interfaces.^{4,15,16} The recombination reduces photocatalytic activity of TiO₂. To increase the photocatalytic activity, the nanosize TiO₂ such as TiO₂ nanotubes can be deposited with noble metal nanoparticles.^{4,15–19} Among the various noble metals, Ag is one of the most promising metals, because Ag nanoparticles can act as electron trappers, facilitating electron–hole separation and promote interfacial electron transfer process. Thus, the Ag nanoparticles can slow the rate of electrons and holes recombination on the surface of TiO₂. In addition, Ag nanoparticles can easily fabricated by a photochemical deposition method under UV light irradiation.¹⁹

These days, as a method in fabricating nanostructures, block copolymer (BCP) lithography which could replace conventional photolithography, has been widely investigated as the nanolithography of various applications such as next generation semiconductors, information storage, photonic crystals, etc.^{20–24} A BCP material consists of two or more polymeric chains joined by a covalent bond, which is chemically different and attached each

other. BCP materials can self-assemble into uniformly ordered nanostructures such as cylindrical, lamellar, spherical, and gyroid morphologies by phase separation because of connectivity constraints and the incompatibility between the two blocks. Differently ordered nanostructures can be fabricated by varying the volume fraction of the two blocks. The lithography using block copolymer thin films as templates has emerged as a promising low cost and high throughput technology. One of the most common block copolymers, polystyrene–block–polymethylmethacrylate (PS–*b*-PMMA) allows selective removal of PMMA domains by wet and dry methods, and remaining PS domains can serve as the mask for pattern transfer.

There are many methods in fabricating nanosize TiO₂, and among these, TiO₂ nanotubes can have higher surface to volume ratio, therefore, higher photocatalytic properties can be expected. In this study, we fabricated uniform TiO₂ nanotubes with 40 nm scale diameter using a self-aligned, hole patterned BCP template lithography, atomic layer deposition of TiO₂, and reactive ion etching (RIE). In addition, we deposited the Ag nanoparticles onto TiO₂ nanotubes using a photochemical deposition method under UV light irradiation. Then, we compared the photocatalytic degradation of methylene blue (MB) dye solution containing TiO₂ and Ag/TiO₂ nanotubes under UV light irradiation to evaluate the photocatalytic effects. The results showed that the photocatalytic effect of Ag/TiO₂ nanotubes was higher than that of TiO₂ nanotubes without Ag nanoparticles demonstrating the Ag nanoparticles on TiO₂ nanotubes as enhanced photocatalysts.

2. EXPERIMENTAL DETAILS

2.1. Formation of BCP Lithographic Hole Array Pattern on Silicon Substrates

In order to form a PS hole array pattern of 40 nm diameter on silicon substrates, first, the silicon substrates were cleaned with acetone, alcohol, and deionized water. Then, a PS–*b*-PMMA solution composed of 140 kg/mol PS and 65 kg/mol PMMA dissolved in toluene was spin-coated on the cleaned silicon substrates and annealed at 230 °C for 40 h in a vacuum chamber for the phase separation of PS and PMMA. After the thermal annealing, 40 nm diameter cylinder array of PMMA was formed in the PS matrix. The PMMA cylinders in PS matrix were selectively removed by an O₂ plasma at room temperature. As a result, a PS hole array pattern of 40 nm diameter and 60 nm height was formed in the silicon substrates. This PS hole array pattern on silicon substrates was used as the lithographic mask for pattern transfer process.

2.2. Fabrication of Ag/TiO₂ Nanotubes

The silicon substrates with the PS hole array mask pattern were etched by a reactive ion etch (RIE) system. The etching process was carried out with 13.56 MHz 50 W power,

45 sccm of SF₆, and 6.5 mTorr of operating pressure for 45 sec. During the etching process, the main chamber's temperature was maintained at room temperature with a chiller because the PS mask was soft and can be easily collapsed at high temperatures. After the silicon etching, the PS mask was selectively removed by the etcher with 13.56 MHz 50 W power, 50 sccm O₂, and 5 mTorr of operating pressure for 1 min to expose 40 nm diameter uniform silicon hole array pattern. Next, 12 nm thick TiO₂ was uniformly deposited on the silicon hole array pattern by atomic layer deposition (ALD) system (SNTEK, SN 13-20, Korea) using the precursors of titanium isopropoxide (TTIP 97%, Sigma-Aldrich Inc., USA) and O₂. In this process, the growth rate of TiO₂ was about 1.36 Å/cycles at 200 °C. When the structure of deposited TiO₂ was investigated by X-ray diffraction, anatase TiO₂ could be identified for ALD TiO₂ thicker than ~100 nm (for 12 nm thick ALD deposited TiO₂, possibly due to the too thin thickness, the phase of the deposited TiO₂ could not be identified). The top and bottom regions (not the sidewall region of the silicon hole array) of the deposited TiO₂ on the silicon hole array pattern were selectively etched by using the RIE system at 13.56 MHz 50 W bias power, 30 sccm Ar, 20 sccm C₂F₆, and 8 mTorr of operating pressure for 1 min 30 sec. After that, the remained silicon attached to TiO₂ nanotubes was also selectively removed by RIE with 13.56 MHz 50 W bias power, 25 sccm SF₆, and 6.5 mTorr of operating pressure for 45 sec to expose TiO₂ nanotubes (TiO₂ remaining on the sidewall of silicon hole array) from the silicon substrates.

On the uniformly fabricated TiO₂ nanotube array, a photochemical Ag deposition process was carried out. First, the TiO₂ nanotube array samples were immersed in a petri dish filled with 1 mM AgNO₃ aqueous solution and sonicated for 30 min. Next, the TiO₂ nanotube array samples in the AgNO₃ (silver nitrate, 99.8%, Daejung Inc., Korea) aqueous solution were irradiated with a 12 W UV lamp. The UV irradiation time was varied from 10 to 30 min. After the photochemical process, the samples were washed by deionized water and dried out in an oven at 25 °C for 2 h.

2.3. Characterization

A scanning electron microscopy (FE-SEM, Hitachi, S-4700, Japan) was used to observe the top, cross-sectional, and tilt views of the PS hole array mask pattern, TiO₂ nanotubes, and Ag/TiO₂ nanotubes. X-ray photoelectron spectroscopy (XPS, Thermo, ESCALAB250, England) was used to analyze the surface binding states of the TiO₂ nanostructures deposited by ALD and Ag/TiO₂ nanotubes. For the photolysis test of a dye material, methylene blue was used. Various TiO₂ and Ag/TiO₂ nanostructure samples were immersed in the petri dishes filled with 5ppm methylene blue (C₁₆H₁₈ClN₃S, Junsei Chemical, Japan) aqueous solution and were covered with quartz

plates. Then, in dark condition, the samples were irradiated with a 100 W UV lamp emitting a wavelength of 254 nm. A UV-visible spectrophotometer (UV/Vis spectrophotometer, Libra, S2, Biochrom, USA) was used for the measurement of periodical absorbance of the methylene blue aqueous solution at the wavelength of 660 nm.

3. RESULTS AND DISCUSSION

Figure 2 shows cross-sectional, top, and tilted SEM images of (a and b) the PS hole array mask pattern formed by BCP lithography on silicon substrates and (c and d) the silicon hole array after etching the silicon with the PS mask and after the PS mask was removed. As shown in Figures 2(a and b), after the BCP phase separation of PS and PMMA, due to the self-assembly, 60 nm height and 40 nm diameter PS hole array mask pattern was formed by the BCP lithography on silicon substrates. As shown in Figures 2(c and d), after the etching of silicon using the PS hole array mask followed by the removal of the PS mask 60 nm depth and 40 nm diameter uniform silicon hole array with the aspect ratio of 1.5 could be fabricated.

To obtain TiO₂ nanotubes, 12 nm thick TiO₂ was deposited on the 40 nm diameter/60 nm height silicon hole array by ALD using titanium isopropoxide and oxygen as precursors. The TiO₂ growth rate was about 1.36 Å/cycle at 200 °C. Figures 3(a and b) show the cross-sectional and tilted SEM images of TiO₂ deposited on silicon hole array by the ALD, respectively. As shown in Figures 3(a and b), uniform TiO₂ with the thickness of about 12 nm was deposited on the all over the silicon surfaces including the silicon hole area. But, as shown in Figures 3(a and b), the silicon hole was not fully filled with TiO₂ leaving a narrow hole in the silicon hole pattern.

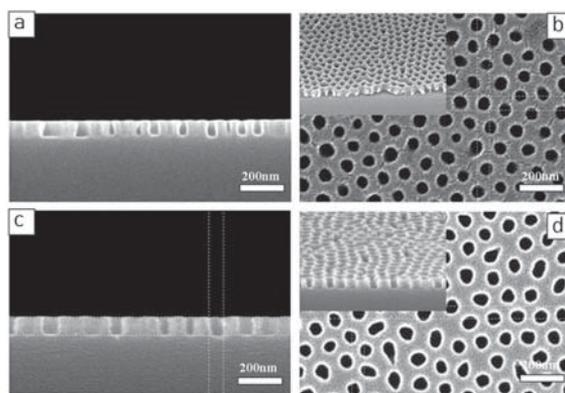


Fig. 2. SEM images of (a) cross-section view and (b) top view of the 60 nm height and 40 nm diameter of the PS hole array mask pattern formed by BCP lithography on silicon substrate. (c and d) are the SEM images of cross-section view and top view of silicon hole array after etching silicon with the PS mask, respectively. After the silicon etching, the silicon hole size was 40 nm and the etch depth was 60 nm. The insets in (b and d) show the SEM images of tilted view of PS hole array mask pattern and etched silicon hole array, respectively.

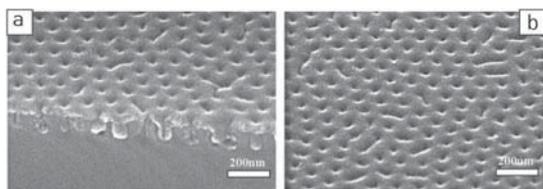


Fig. 3. (a) Cross-sectional and (b) tilted SEM images of TiO₂ deposited on silicon hole array by ALD with titanium isopropoxide and O₂ precursors. The TiO₂ growth rate was about 1.36 Å/cycle at 200 °C and the TiO₂ thickness was about 12 nm.

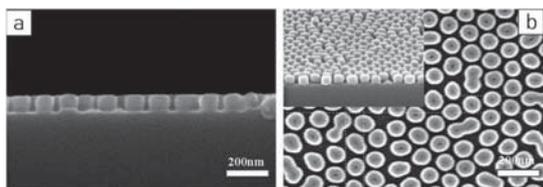


Fig. 4. SEM images (a) cross-sectional, (b) top, and inset of (b) tilted views of fabricated TiO₂ nanotubes.

To form TiO₂ nanotubes, TiO₂ in the top and bottom areas of silicon hole array was selectively etched, therefore, only TiO₂ was remaining on the sidewall of the silicon hole array. After that, the silicon attached with sidewall TiO₂ was selectively removed to obtain TiO₂ nanotubes. Figure 4 shows SEM images (a: cross-sectional, b: top, and inset of (b): tilted views) of fabricated TiO₂ nanotubes. As shown in the figure, after the silicon removal, 50~60 nm height and 40~50 nm diameter TiO₂ nanotubes could be obtained on the silicon substrate.

On the fabricated TiO₂ nanotubes, to investigate the effect of Ag nanoparticles attached on the TiO₂ nanotubes, TiO₂ nanotubes were immersed in a 1 mM AgNO₃ solution and were irradiated with 12 W UV light for 10~30 min for the nucleation of Ag nanoparticles on TiO₂ nanotube surfaces. Figure 5 shows the SEM images of

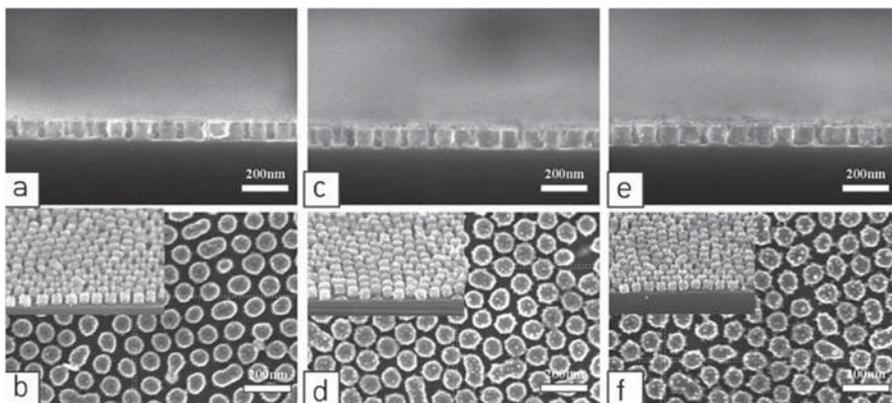


Fig. 5. SEM images (cross-sectional, top, and tilted views) of Ag deposited TiO₂ nanotubes using a photochemical deposition method. To deposit Ag nanoparticles, TiO₂ nanotubes were immersed in a 1 mM AgNO₃ solution and were irradiated with 12 W UV light for (a, b) 10 min, (c, d) 20 min, and (e, f) 30 min. The average size of formed Ag nanoparticles was about 7~9 nm.

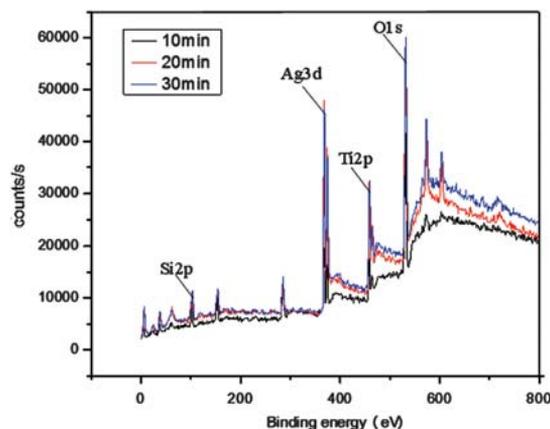


Fig. 6. XPS wide scan data of TiO₂ nanotubes deposited with Ag nanoparticles for 10, 20, and 30 min.

TiO₂ nanotubes deposited with Ag nanoparticles by the immersion in the 1 mM AgNO₃ solution while irradiating with 12 W UV light for (a, b) 10 min, (c, d) 20 min, and (e, f) 30 min. As shown in Figure 5, the precipitation of Ag nanoparticles on the TiO₂ nanotube surface could be observed by the immersion in the AgNO₃ solution and, with the increase of immersion time, the number of Ag nanoparticles on the TiO₂ surfaces was increased. The measured size of the Ag nanoparticles was in the range from 7 to 9 nm. For the further identification of the formation of Ag nanoparticles on TiO₂ nanotubes after the immersion in the AgNO₃ solution, XPS wide scan data were observed for the TiO₂ nanotubes immersed in the AgNO₃ solution while irradiating with 12 W UV light for 10~30 min and the results are shown in Figure 6. As shown in Figure 6, with the increase of UV irradiation time, the increased Ag peak intensities were observed indicating the increased number of Ag nanoparticles.

Using the TiO₂ nanotubes with and without Ag nanoparticles, the photocatalytic properties were investigated. For the photocatalytic properties, the samples were immersed in a 5 ppm, 15 ml methylene blue aqueous solution and irradiated with a 100 W UV lamp emitting a wavelength of 254 nm up to 3 hours. The change of concentration by the decomposition of the methylene blue solution was measured by measuring the change of light absorption at 660 nm, which is the peak absorption wavelength of the methylene blue, using a UV-visible spectrometer. Figure 7 shows the change in the concentration of the methylene blue aqueous solution for various TiO₂ samples. As references, the changes of the concentration with the UV radiation for the methylene blue aqueous solution without TiO₂ and for the 12 nm thick TiO₂ thin film deposited on flat silicon substrate were included in addition to the concentration changes for the TiO₂ nanotubes and TiO₂ nanotubes deposited with Ag nanoparticles for 10, 20, and 30 min. As shown in Figure 7, no noticeable change of concentration was observed for the methylene blue solution without TiO₂ while the solution with 12 nm TiO₂ flat thin film showed about 2% of concentration change after 3 h. The solution with the TiO₂ nanotubes exhibited higher concentration change of the methylene blue solution possibly due to the increased surface area. The TiO₂ nanotubes with Ag nanoparticles exhibited the higher concentration change compared to TiO₂ nanotubes alone by facilitating electron-hole separation and promoting interfacial electron transfer process through the Ag nanoparticles. Among the TiO₂ nanotubes with Ag nanoparticles, the Ag nanoparticles deposited for 20 min exhibited the most prominent photocatalytic effect among the TiO₂ with Ag nanoparticles by reducing the concentration of the methylene blue aqueous solution about 8% after 3 h of UV radiation.

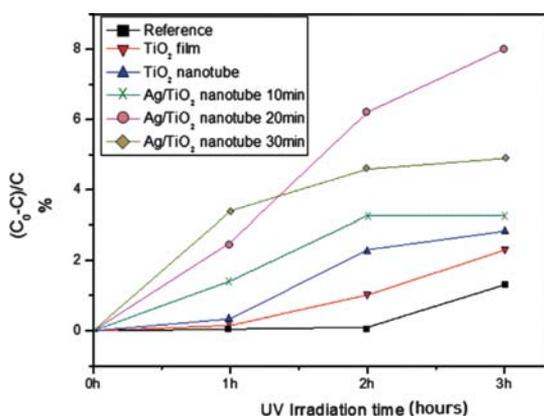


Fig. 7. Change in the concentration of a 5 ppm methylene blue aqueous solution with 100 W UV light for various TiO₂ samples. As references, the change of the concentration with UV radiation for the methylene blue aqueous solution without TiO₂ and for the 12 nm thick TiO₂ thin film deposited on flat silicon substrate were included in addition to the TiO₂ nanotubes and TiO₂ nanotubes deposited with Ag nanoparticles for 10, 20, and 30 min.

The lower photocatalytic effect for the TiO₂ deposited with Ag nanoparticles for 30 min appears to be related to the decreased UV penetration to TiO₂ by the increased surface area covered with Ag nanoparticles.

4. CONCLUSION

Using the TiO₂ nanotubes fabricated with BCP lithography, TiO₂ ALD, and etching and Ag nanoparticles precipitated on TiO₂ nanotubes by a photochemical method, the photocatalytic effect of TiO₂ nanotubes with/without Ag nanoparticles was investigated by measuring the concentration change of a methylene blue solution. The photocatalytic effect of TiO₂ nanotubes was higher than flat TiO₂ film because of the increased TiO₂ surface area. Also, Ag nanoparticles attached on the TiO₂ nanotubes further improved the photocatalytic effect by facilitating electron-hole separation and promoting interfacial electron transfer process through the Ag nanoparticles. However, when the amount of Ag nanoparticles on TiO₂ nanotubes are too much, possibly due to the decreased UV penetration to TiO₂ by the increased surface area covered with Ag nanoparticles, the photocatalytic effect was decreased. By precipitating 7~9 nm size Ag nanoparticles for 20 min on 40~50 nm diameter/50~60 nm height TiO₂ nanotubes, the highest photocatalytic effect could be obtained.

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References and Notes

1. A. Word, *Chem. Mater.* 5, 280 (1993).
2. D. Wang, Z. H. Zhou, H. Yang, K. B. Shen, Y. Huang, and S. Shen, *R. Soc. P.* 22, 16306 (2012).
3. R. Liu, P. Wang, X. Wang, H. Yu, and J. Yu, *Phys. Chem. B* 116, 17721 (2012).
4. H. M. Sung-Suh, J. R. Choi, H. J. Hah, S. M. Koo, and Y. C. Bae, *J. Photochem. Photobiol. A* 163, 37 (2003).
5. M. Paulose, K. Shankar, O. K. Varghese, G. K. Mor, and C. A. Grimes, *J. Phys. D: Appl. Phys.* 39, 2498 (2006).
6. J. Keleher, J. Bashant, N. Heldt, L. Johnson, and Y. Li, *World J. Microbiol. Biotechnol.* 18, 133 (2002).
7. Y. Lai, H. Zhuang, K. Xie, D. Gong, Y. Tang, L. Sun, C. Lin, and Z. Chen, *New J. Chem.* 34, 1335 (2010).
8. J. M. Macak, H. Tsuchiya, A. Ghicov, K. Yasuda, R. Hahn, S. Bauer, and P. Schmuki, *Curr. Opin. Solid State Mater. Sci.* 11, 3 (2007).
9. Y. Liao, H. Zhang, Z. Zhong, L. Jia, F. Bai, J. Li, P. Zhong, H. Chen, and J. Zhang, *ACS Appl. Mater. Inter.* 5, 11022 (2013).
10. J. Schneider, M. Matsuoka, M. Takeuchi, J. Zhang, Y. Horiuchi, M. Anpo, and D. W. Bahnemann, *Chem. Rev.* 114, 9919 (2014).
11. C. H. Wu and J. M. Chern, *Ind. Eng. Chem. Res.* 45, 6450 (2006).
12. I. K. Konstantinou and T. A. Albanis, *Appl. Catal. B-Environ.* 49, 1 (2004).
13. D. O. Scanlon, C. W. Dunnill, J. Buckeridge, S. A. Shevlin, A. J. Logsdail, S. M. Woodley, C. R. A. Catlow, M. J. Powell, R. G.

- Palgrave, I. P. Parkin, G. W. Watson, T. W. Keal, P. Sherwood, A. Walsh, and A. Sokol, *Nat. Mater.* 12, 798 (2013).
14. K. Woan, G. Pyrgiotakis, and W. Sigmund, *Adv. Mater.* 21, 2233 (2009).
15. T. T. Y. Tan, C. K. Yip, D. Beydoun, and R. Amal, *Chem. Eng. J.* 95, 179 (2003).
16. Y. C. Liang, C. C. Wang, C. C. Kei, Y. C. Hsueh, W. H. Cho, and T. P. Perng, *J. Phys. Chem. C* 115, 9498 (2011).
17. I. Paramasivam, J. M. Macak, A. Ghicov, and P. Schmuki, *Chem. Phys. Lett.* 445, 223 (2007).
18. L. Miao, Y. Ina, S. Tanemura, T. Jiang, M. Tanemura, K. Kaneko, S. Toh, and Y. Mori, *Surf. Sci.* 601, 2792 (2007).
19. Y. Ohko, T. Tatsuma, T. Fujii, K. Naoi, C. Niwa, Y. Kumota, and A. Fujishima, *Nat. Mater.* 2, 29 (2003).
20. Z. Liu, X. Gu, J. Hwu, S. Sassolini, and D. L. Olynick, *Nanotechnology* 25, 285301 (2014).
21. S. W. Choi, J. H. Shin, M. H. Jeon, J. H. Mun, S. O. Kim, G. Y. Yeom, and K. N. Kim, *J. Nanosci. Nanotechnol.* 15, 8093 (2015).
22. J. Y. Cheng, C. A. Ross, E. L. Thomas, H. I. Smith, and G. J. Vancso, *Appl. Phys. Lett.* 81, 3657 (2002).
23. X. Gu, Z. Liu, I. Gunkel, S. T. Chourou, S. W. Hong, D. L. Olynick, and T. P. Russell, *Adv. Mater.* 24, 5688 (2012).
24. S. O. Kim, H. H. Solak, M. P. Stoykovich, N. J. Ferrier, J. J. D. Pablo, and P. F. Nealey, *Nature* 424, 411 (2003).

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